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A New Way to Measure Low Molecular Weight Polymer**John H Jennings**

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DOI: <https://doi.org/10.62225/2583049X.2024.4.2.2492>Corresponding Author: **John H Jennings****Abstract**

Grubisic, *et al* (1967) ^[1] used the Einstein viscosity law (1906) ^[2] as a theoretical basis in calibrating a column for size exclusion chromatography (SEC) and it works well for polymer molecular weight (MW) from 10,000 to 11 million daltons. There are problems measuring polymer MW between 500 and 5000 daltons with SEC. For instance, Chance, *et al* (1995) ^[3] say that SEC for very low molecular weight suffers from intrinsic viscosities becoming negative for certain polymers due to packing effects. Also, how are the pores going to distinguish small polymers? Modern light

scattering detectors, Podzimek (2014) ^[4], can effectively characterize polymers down to a several hundred daltons. Jennings (2012) ^[5] published an equation for bubble nucleation in polymer solutions that fairly well agrees with Jennings/Middleman data (1985) ^[6] and is a new way to measure polymer MW from polystyrene (PS) dimer up to about 5000 daltons. Jennings/Middleman used cyclohexane as the solvent and in this paper, it is easy to show how their equation can estimate low molecular weight polymers and details of the derivation will be shared.

Keywords: Einstein Viscosity Law, Size Exclusion Chromatography, Bubble Nucleation, Polymer Solutions, Light Scattering, Low Molecular Weight Polymer

Introduction

Here, the author is discussing his way of measuring low molecular weight polymer. SEC is a way to measure high polymer but fails when applied to low molecular weight polymer. One good way to characterize low MW polymer is light scattering and this is described in Podzimek. Jennings (2012) published his theory for measuring the limit of superheat of polymer solutions and this is outlined in the Results section.

Results

$$\partial T / \partial w_2 = (MW_1 / MW_2) (3kTl^2 / \sigma_1 a) \quad (18)$$

Eq. (18) is from my June 4, 2012 paper in International Journal of Thermodynamics (ISTANBUL, Turkey). Look at Fig 2 from the Macromolecules paper at the end to see the slopes for 2000 and 4000 MW. The slopes are $\partial T / \partial w_2$, so you solve it this way. I can propose that this equation is a fairly accurate way of measuring molecular weight from 500 to 5000 daltons. Water (H₂O) is 18 daltons.

$$MW_2 = (MW_1) (3kTl^2 / \sigma_1 a) / (\partial T / \partial w_2) \quad (18) \text{ solved for } MW_2$$

MW₂ desired quantity = molecular weight of polymer

MW₁ molecular weight of solvent

K Boltzmann constant

T temperature of polymer solution

l₁ limit of superheat of solvent

σ₁ surface tension of solvent at l₁

a surface area of solvent molecule at l₁

w₂ weight fraction polymer

r degree of polymerization

In my experiment, cyclohexane was the solvent and the polymer was polystyrene. Second to the end, is the image of the apparatus used to measure the limit of superheat. Jennings/Middleman did not use Argon and the spacing of the Nichrome wire was closer and closer toward the top of the column to allow for a gradient in temperature: Holden/Katz (1978)^[7] is the source for the apparatus image.

Discussion

In the original theory article by Jennings (2012)^[5], he was thorough in taking $d\ln A/dK$ in both the w direction and T direction, which are orthogonal. In the w direction, it was simple, but for the T direction there obtained a lengthy expression that gave the same result, about -0.2% . Here from International Journal of Thermodynamics are the two results.

$$\lim_{w_2 \rightarrow 0} (\partial \ln A / \partial w_2) / (\partial K / \partial w_2) = 1/(6K) = -0.24\% \quad (A)$$

$$\lim_{T \rightarrow T_l} (\partial \ln A / \partial T) / (\partial K / \partial T) = \text{complicated expression} = -0.19\% \quad (B)$$

The reason (A) and (B) do not agree exactly has got to be in the estimate for the surface tension.

For the nucleation rate in polymer solutions, the surface tension has the most influence. The details of the calculation for Eq. (18) above are all laid out in Jennings (2012)^[5].

Eq. (18) had its beginning in 2012 and it wasn't until Jennings (2023)^[8] that it was proved for dimer ($r = 2$) up to high polymer ($r = \text{infinity}$). Here we announce that using the rising drop method, it should be possible to measure low molecular weight polymer inexpensively. Notice that the limit of superheat rise for 2000 MW and 4000 MW is of the order of 10 to 15 degrees Centigrade allowing for a rather exact estimate of MW². The data for bubble nucleation in polymer solutions first appeared in Jennings/Middleman and they have the definitive solution to the problem.

Conclusions

Theoretical work by Blander/Katz (1975)^[9] and Siow/Patterson (1973)^[10] were combined in Jennings (2012)^[5] to produce Eq. (18) above. In this paper, the author reveals the rationale for using Eq. (18) to measure low molecular weight polymer and it is a general equation.

Acknowledgments

For my part, I'm moved to mention Father Ivan Tou at Newman Hall, who has supported me in his ministry there. Even though Father Ivan is moving on, the congregation has benefited from him. Father Ivan definitely supported my science efforts.

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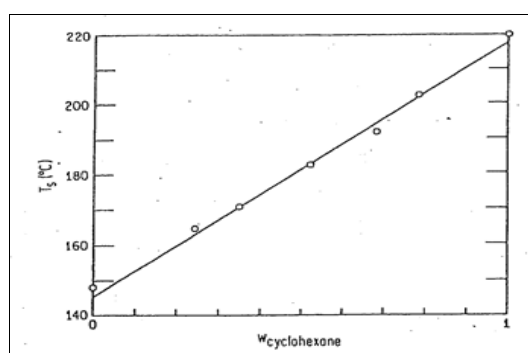


Fig 1: Data on limiting superheat for binary solution of cyclohexane and pentane. Composition is mass fraction

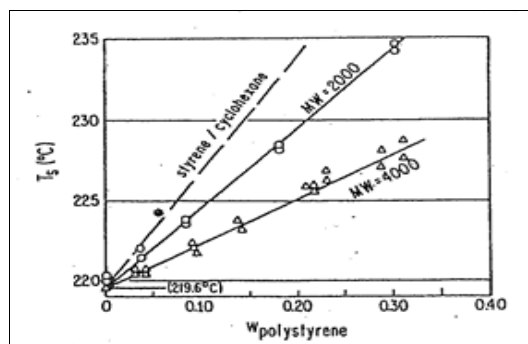


Fig 2: Data on limiting superheat for low molecular weight polystyrene in cyclohexane. T_s for pure cyclohexane is taken as 219.6 °C

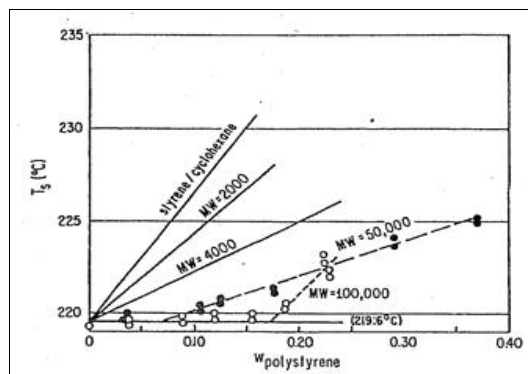


Fig 3: Data on limiting superheat for high molecular weight polystyrene in cyclohexane

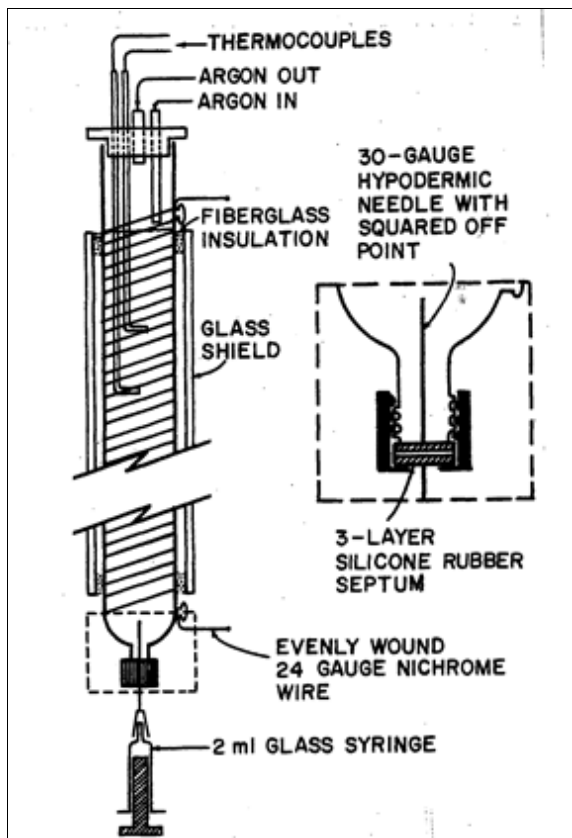


Fig 4: Low gradient rising droplet column and detail